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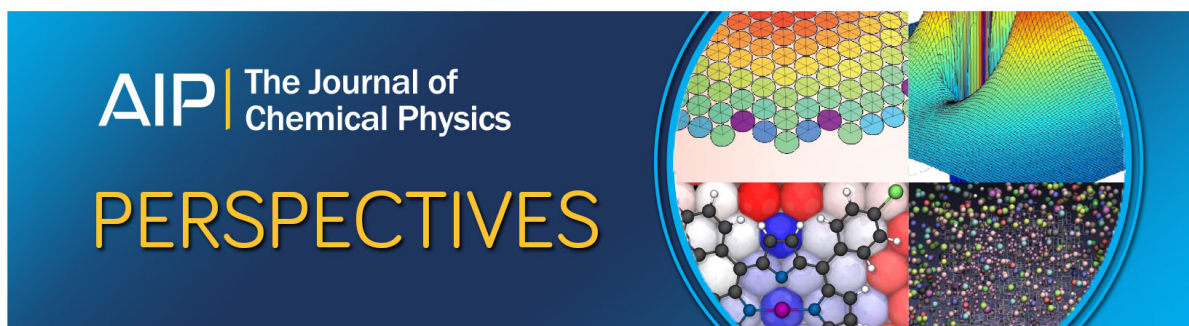
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Self-consistent second-order Green's function perturbation theory for periodic systems

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Despite recent advances, systematic quantitative treatment of the electron correlation problem in extended systems remains a formidable task. Systematically improvable Green's function methods capable of quantitatively describing weak and at least qualitatively strong correlations appear as promising candidates for computational treatment of periodic systems. We present a periodic implementation of temperature-dependent self-consistent 2nd-order Green's function (GF2) method, where the self-energy is evaluated in the basis of atomic orbitals. Evaluating the real-space self-energy in atomic orbitals and solving the Dyson equation in \mathbf{k} -space are the key components of a computationally feasible algorithm. We apply this technique to the one-dimensional hydrogen lattice — a prototypical crystalline system with a realistic Hamiltonian. By analyzing the behavior of the spectral functions, natural occupations, and self-energies, we claim that GF2 is able to recover metallic, band insulating, and at least qualitatively Mott regimes. We observe that the iterative nature of GF2 is essential to the emergence of the metallic and Mott phases. © 2016 AIP Publishing LLC. [<http://dx.doi.org/10.1063/1.4940900>]

I. INTRODUCTION

In all-electron solid state calculations, density functional theory (DFT) is enormously successful.^{1–11} Despite its popularity, however, certain problems are persistently hard to address, such as prediction of atomization energies and heats of formation,¹² band gaps and optical properties,^{11–15} as well as van der Waals interactions.^{12,16,17} Though new classes of exchange-correlation functionals are often able to offer superior performance, systematic improvability of DFT remains an unresolved issue.

Ab initio methods offer a systematic improvement, starting from a mean-field theory such as Hartree–Fock (HF)^{18–20} and progressing to the hierarchies of perturbation,²¹ coupled-cluster (CC),²² and configuration interaction (CI)²³ theories. These methods, however, are prohibitively costly when applied to crystalline systems, and while excellent in describing molecular systems, they may face additional challenges when applied to periodic problems.²⁴ Though overcoming some of these impediments requires technical advancement including adaptation to a finite-temperature formalism to enable prediction of competing electronic phases present at finite temperatures, certain challenges, such as divergencies for metals in 2nd-order Møller–Plesset perturbation theory (MP2) and the problem of both MP2 and CC in describing a multi-reference character of the electronic states in phases such as Mott insulator, remain more ideological. We would like to stress that these problems are not insurmountable, and promising approaches have recently emerged;²⁵ nevertheless, to successfully conquer the physics of periodic systems, it may require significant ideological as well as implementation efforts.

Traditional *ab initio* methods such as HF,^{26–29} MP2,^{27–41} coupled cluster singles and doubles (CCSD),^{28,29,42,43} or the method of increments,^{44–50} which divides the correlation energy into contributions of localized orbitals, recover structural and energetic properties such as cohesive (atomization) energies, lattice constants, bulk moduli, unit cell energies, and band gaps. However, these methods, while excellent in recovering electronic energy and some simple spectral properties like the magnitude of the band gap, are not designed to produce the \mathbf{k} -dependent density of states — a property measured in angle-resolved photoelectron spectroscopy (ARPES) that remains as one of the most popular experimental techniques for studying crystalline systems. Moreover, these methods do not yield the self-energy that allows us to distinguish band and Mott insulators and gives additional information about the nature of correlations present in the system under consideration.

Recently, a lot of progress has been made in numerical Green's function methods, and currently, efficient implementations of zero-temperature random phase approximation (RPA)^{51,52} and GW⁵³ can be applied to solids. These methods traditionally focus on the evaluation of spectral properties such as the experimentally observed density of states. Yet again, similarly to the *ab initio* methods, they cannot properly illustrate metal-to-insulator transitions and phases that display strong correlations.

Periodic systems are very challenging for quantum chemical/condensed matter methods, and clearly this territory is much more uncharted than the molecular realm. Consequently, there is plenty of room for methodological and computational improvement. In this paper, we investigate a periodic implementation of temperature-dependent 2nd-order iterative Green's function (GF2) method.^{54–56} To learn about the performance of GF2 for solids, we analyze a

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one-dimensional (1D) infinite hydrogen lattice and examine the following aspects: (i) the existence of multiple phases at finite temperature, (ii) the ability of GF2 to describe a metallic phase, (iii) the description of metal-to-insulator transition and the Mott phase, and (iv) the change in the solutions due to the iterative procedure. We expect that the finite temperature iterative perturbative approach is not affected by divergencies for metals in the same way as the zero-temperature perturbation theory such as MP2. Additionally, we expect that the iterative procedure present in GF2 is able to account for a range of correlation effects that cannot be described by a non-iterative method.

This paper is organized as follows. In Sec. II, we discuss the advantages of iterative Green's function methods. In Sec. III, we outline the basic components of periodic GF2 and elucidate the local character of the real-space self-energy in the atomic basis. In Sec. IV, we elaborate on the details of the self-consistent procedure. The evaluation of the reciprocal space \mathbf{k} -dependent spectral function is explained in Sec. V. The results of our investigations for the 1D infinite hydrogen lattice are presented in Sec. VI. Finally, we form the conclusions in Sec. VII.

II. ITERATIVE GREEN'S FUNCTION METHODS

The system in thermal equilibrium can be described by a temperature-dependent one-body Green's function $\mathbf{G}(\omega_n)$ on the imaginary axis, where $\omega_n = i(2n + 1)\pi/\beta$ are Matsubara frequencies, $\beta = 1/(k_B T)$ is the inverse temperature, and k_B is the Boltzmann constant. The exact expectation values of any one-body operator can be evaluated using the explicitly temperature-dependent density matrix $\hat{\gamma} = e^{-\beta(\hat{H} - \mu\hat{N})}/\text{Tr}(e^{-\beta(\hat{H} - \mu\hat{N})})$ where \hat{H} , \hat{N} , and μ are the system Hamiltonian, particle number operator, and chemical potential, respectively.

In this paper, we discuss an iterative perturbative 2nd-order approximation, GF2,⁵⁴⁻⁵⁶ yielding an approximate temperature-dependent density matrix from the 2nd-order Green's function. In such method, the perturbation is not only a function of interactions but also temperature. Therefore, GF2 is exact in the limit of infinite temperature as well as vanishing perturbation. Recently, numerous papers appeared that investigate different variants of temperature-dependent perturbation theories,^{27,28} since the inclusion of temperature, while not very relevant in molecules, is crucial in materials studies.

The iterative temperature-dependent Green's function methods such as GF2 are conserving approximations,^{57,58} and the observables obtained from the Green's function agree with the macroscopic conservation laws, e.g., conservation of particle number, momentum, angular momentum, and energy. In such approximations, the self-energy can be expressed as a functional derivative of a certain functional with respect to the Green's function. This is only attained if the Green's function method is self-consistent.⁵⁴ If the Green's function method is not self-consistent, the particle number can be incorrect and the virial theorem does not need to be preserved. Most importantly, in the conserving approximations at self-consistency, different methods for calculating the total energy via the Green's function give the same result.⁵⁹

Another advantage of self-consistency is due to effective inclusion of higher-order diagrams, which are not explicitly present in the 2nd-order self-energy approximation, via iterative renormalization of the free propagator lines. In Sec. VI, we discuss how the iterations (and implicit inclusion of higher-order diagrams) influence the phase diagram and convergence of GF2 for metallic systems.

III. EVALUATING A GREEN'S FUNCTION AND SELF-ENERGY IN THE ATOMIC BASIS

In this section, we discuss the strategy of building a self-consistent 2nd-order Green's function for infinite crystalline species. To this end, we aim to simultaneously take advantage of the following useful representations of the Green's function and self-energy. First, in a fashion similar to Refs. 55 and 56, we are exploiting a convenient denominator-free form of the 2nd-order self-energy in the imaginary-time domain. Then, since the 2nd-order self-energy expression can be evaluated using atomic orbitals, we take advantage of the local character of the self-energy in this basis. Third, to facilitate an efficient evaluation of the Green's function, we construct it and solve the Dyson equation in \mathbf{k} -space.

A. The Green's function and self-energy for periodic systems: Dyson equation in the frequency domain

For an infinite crystalline system, the evaluation of various quantities such as overlap, density and Fock matrices, or Green's functions is made feasible by exploiting translational symmetry and imposing periodic boundary conditions (PBCs). If i, j denote the indices of the functions in the atomic orbital (AO) basis, vectors indices \mathbf{g}, \mathbf{h} enumerate the real-space unit cells, and $\mathbf{0}$ is a chosen (reference) cell, then periodicity of an arbitrary real-space matrix quantity \mathbf{A} implies $A_{ij}^{\mathbf{gh}} = A_{ij}^{\mathbf{0}, \mathbf{h}-\mathbf{g}}$, that is, matrices resulting from a shift by an arbitrary lattice translation vector are identical. From a computational standpoint, it means that the row of matrices $\mathbf{A}^{\mathbf{0g}}$, where \mathbf{g} runs over every crystalline unit cell, contains the entire information about \mathbf{A} .

In the frequency domain, the general expression for the Green's function reads

$$\mathbf{G}(\omega) = [(\omega + \mu)\mathbf{S} - \mathbf{F} - \Sigma(\omega)]^{-1}, \quad (1)$$

where \mathbf{S} and \mathbf{F} , and $\Sigma(\omega)$ are, respectively, overlap and Fock matrices and the self-energy in the AO basis. In this paper, we use a discrete set of Matsubara (imaginary) frequencies ω_n . Eq. (1) can be written as the Dyson equation

$$\mathbf{G}^{-1}(\omega) = \mathbf{G}_0^{-1}(\omega) - \Sigma(\omega), \quad (2)$$

where $\mathbf{G}_0(\omega) = [(\omega + \mu)\mathbf{S} - \mathbf{F}]^{-1}$ serves as an "unperturbed" 0th-order Green's function.

For a crystalline system, Eq. (2) turns into the following expression which facilitates the evaluation of $[\mathbf{G}^{-1}(\omega)]^{\mathbf{0g}}$ in the AO basis in the real space:

$$[\mathbf{G}^{-1}(\omega)]^{\mathbf{0g}} = (\omega + \mu)\mathbf{S}^{\mathbf{0g}} - \mathbf{F}^{\mathbf{0g}} - \Sigma^{\mathbf{0g}}(\omega). \quad (3)$$

Retrieving $[\mathbf{G}(\omega)]^{\mathbf{0g}}$, however, is not attainable by the brute force inversion of the above equation, as each block would

result from the inversion of the entire matrix. A similar problem arises in periodic HF or DFT calculations when the Fock matrix is diagonalized. Due to translational symmetry of the Green's function $\mathbf{G}^{0\mathbf{g}}(\omega) = \mathbf{G}^{\mathbf{h},\mathbf{g}+\mathbf{h}}(\omega)$ though, the real-space inversion can be traditionally circumvented via a Fourier transform to \mathbf{k} -space. This is equivalent to symmetrization of the AO basis (see Ref. 1 for details). In \mathbf{k} -space, the translationally invariant matrices appear in a convenient block-diagonal form,

$$\mathbf{A}^{\mathbf{k}} = \sum_{\mathbf{g}} \mathbf{A}^{0\mathbf{g}} \exp(i\mathbf{k} \cdot \mathbf{g}). \quad (4)$$

Then solving the Dyson equation breaks down into a series of independent $N \times N$ matrix inversions at each \mathbf{k} -point sampling the first Brillouin zone, where N is the number of the AO basis functions in the unit cell,

$$\mathbf{G}^{\mathbf{k}}(\omega) = [(\omega + \mu)\mathbf{S}^{\mathbf{k}} - \mathbf{F}^{\mathbf{k}} - \mathbf{\Sigma}^{\mathbf{k}}(\omega)]^{-1}. \quad (5)$$

The real-space Green's function can be then restored by inverse Fourier transform,

$$\mathbf{G}^{0\mathbf{g}}(\omega) = \frac{1}{V_{BZ}} \sum_{\mathbf{k}} \mathbf{G}^{\mathbf{k}}(\omega) \exp(-i\mathbf{k} \cdot \mathbf{g}), \quad (6)$$

where V_{BZ} is the volume of the first Brillouin zone.

B. Employing the real-space self-energy locality argument

A practical way to obtain $\mathbf{\Sigma}^{0\mathbf{g}}(\omega)$, necessary to solve Eq. (3), involves two steps: $\mathbf{\Sigma}^{0\mathbf{g}}(\tau)$ is produced in the imaginary-time domain and then Fourier transformed from τ to ω domain.

In the imaginary-time domain, the 2nd-order self-energy can be compactly written as a contraction of the Green's function and two-electron integrals (for simplicity, we discuss a closed-shell case which can be generalized to open-shell systems),⁵⁵

$$\begin{aligned} \Sigma_{ij}(\tau) = & - \sum_{klmnpq} G_{kl}(\tau) G_{mn}(\tau) G_{pq}(-\tau) \\ & \times v_{imqk} (2v_{jnpl} - v_{jlpn}), \end{aligned} \quad (7)$$

where v_{ijkl} are the two-electron integrals in chemists' (11|22) notation (we use this notation through the rest of the text).

A generalization of this expression to the periodic case requires the following real-space blocks of $\mathbf{\Sigma}(\tau)$ to be evaluated:

$$\begin{aligned} \Sigma_{ij}^{0\mathbf{g}}(\tau) = & - \sum_{\mathbf{g}_1, \dots, \mathbf{g}_6} \sum_{klmnpq} G_{kl}^{\mathbf{g}_1\mathbf{g}_6}(\tau) G_{mn}^{\mathbf{g}_1\mathbf{g}_4}(\tau) G_{pq}^{\mathbf{g}_5\mathbf{g}_2}(-\tau) \\ & \times v_{imqk}^{0\mathbf{g}_1\mathbf{g}_2\mathbf{g}_3} (2v_{jnpl}^{\mathbf{g}_4\mathbf{g}_5\mathbf{g}_6} - v_{jlpn}^{\mathbf{g}_6\mathbf{g}_5\mathbf{g}_4}). \end{aligned} \quad (8)$$

In this expression, the bold indices $\mathbf{0}, \mathbf{g}, \mathbf{g}_1, \dots, \mathbf{g}_6$ are the lattice translation vectors pointing to the unit cell where the basis function is located, and $\mathbf{0}$ is the reference cell.

In Eq. (7), the evaluation of the formally 6-fold contractions over the basis functions is facilitated by breaking it down into several consecutive transformations. For instance, the following intermediate contractions can be utilized: $A_{imql} = \sum_k G_{kl}(\tau) v_{imqk}$, $B_{inql} = \sum_m G_{mn}(\tau) A_{imql}$, C_{inpl}

$= \sum_m G_{pq}(-\tau) B_{inql}$, and finally $\Sigma_{ij}(\tau) = - \sum_{npl} C_{inpl} (2v_{jnpl} - v_{jlpn})$, thus bringing the computation cost to $O(N^5)$ (we omit cell indices for simplicity). In the periodic case, we need to produce $\Sigma_{ij}^{0\mathbf{g}}(\tau)$ for every cell index \mathbf{g} with index i pointing to the $\mathbf{0}$ cell only, thus keeping the cost of the evaluation manageable. In addition, screening the two-electron integrals and discarding negligible intermediate contractions can reduce the computation cost even further.

The self-energy expression (Eq. (8)) appears as a product of quantities that decay with the increase of the intercell distance. Such structure of $\Sigma_{ij}^{0\mathbf{g}}(\tau)$ prompts a higher decay rate of the self-energy than that of the individual $G(\tau)$ and v components as the distance between the cells increases. This means that the self-energy is a relatively local quantity despite the summations in Eq. (8) covering the entire crystal.

To demonstrate the local nature of the self-energy, we sketch a short numerical analysis for the 1D equidistant hydrogen lattice with one s -function per atom. Let us closely examine one of the terms in Eq. (8),

$$G_{kl}^{\mathbf{g}_3\mathbf{g}_6}(\tau) G_{mn}^{\mathbf{g}_1\mathbf{g}_4}(\tau) G_{pq}^{\mathbf{g}_5\mathbf{g}_2}(-\tau) v_{imqk}^{0\mathbf{g}_1\mathbf{g}_2\mathbf{g}_3} v_{jnpl}^{\mathbf{g}_4\mathbf{g}_5\mathbf{g}_6}. \quad (9)$$

This term can be considered in two limits: large values of the (i) two-electron integrals or (ii) Green's functions.

(i) *Large integrals limit.* In the chosen basis, the largest integrals are the on-site ones $v_{jjjj}^{\mathbf{g}\mathbf{g}\mathbf{g}\mathbf{g}}$. The corresponding absolute value of the term (9) is $|G_{ij}^{0\mathbf{g}}(\tau) G_{ij}^{0\mathbf{g}}(\tau) G_{ji}^{\mathbf{g}\mathbf{0}}(-\tau) v_{iiii}^{0\mathbf{0}\mathbf{0}\mathbf{0}} v_{jjjj}^{\mathbf{g}\mathbf{g}\mathbf{g}\mathbf{g}}|$. Since $v_{iiii}^{0\mathbf{0}\mathbf{0}\mathbf{0}} v_{jjjj}^{\mathbf{g}\mathbf{g}\mathbf{g}\mathbf{g}} = v_{iiii}^{0\mathbf{0}\mathbf{0}\mathbf{0}} v_{jjjj}^{0\mathbf{0}\mathbf{0}\mathbf{0}}$, the whole expression can be written as

$$|G_{ij}^{0\mathbf{g}}(\tau) G_{ij}^{0\mathbf{g}}(\tau) G_{ji}^{\mathbf{g}\mathbf{0}}(-\tau) v_{iiii}^{0\mathbf{0}\mathbf{0}\mathbf{0}} v_{jjjj}^{0\mathbf{0}\mathbf{0}\mathbf{0}}|. \quad (10)$$

The magnitude of the above expression depends on the value of the triple Green's functions product. In Fig. 1, we plot a typical behavior of $G_{ij}^{0\mathbf{g}}(\tau)$ with respect to \mathbf{g} for several values of the interatomic spacing parameter R . Although, especially in the short bond regime ($R = 0.75 \text{ \AA}$), the Green's function assumes non-negligible values within 10 unit cells, the triple product remains of the orders of 10^{-6} for G_{11} and 10^{-4} for G_{12} . Therefore, for the largest values of the two-electron integrals, the product of the Green's functions $G_{ij}^{0\mathbf{g}}$ ensures a rapid decay of the corresponding contribution to the self-energy.

(ii) *Large Green's function limit.* We maximize the Green's function elements in expression (9) by choosing the on-site elements: $G_{kk}^{0\mathbf{0}}(\tau) G_{mm}^{0\mathbf{0}}(\tau) G_{pp}^{0\mathbf{0}}(-\tau)$. Consequently, expression (9) reduces to

$$G_{kk}^{0\mathbf{0}}(\tau) G_{mm}^{0\mathbf{0}}(\tau) G_{pp}^{0\mathbf{0}}(-\tau) v_{impk}^{0\mathbf{0}\mathbf{0}\mathbf{0}} v_{jmpk}^{\mathbf{g}\mathbf{0}\mathbf{0}\mathbf{0}}. \quad (11)$$

If we further maximize this expression by setting $v_{impk}^{0\mathbf{0}\mathbf{0}\mathbf{0}}$ to its largest value $v_{iiii}^{0\mathbf{0}\mathbf{0}\mathbf{0}}$, we obtain

$$G_{ii}^{0\mathbf{0}}(\tau) G_{ii}^{0\mathbf{0}}(\tau) G_{ii}^{0\mathbf{0}}(-\tau) v_{iiii}^{0\mathbf{0}\mathbf{0}\mathbf{0}} v_{jjii}^{\mathbf{g}\mathbf{0}\mathbf{0}\mathbf{0}}. \quad (12)$$

The integral $v_{jjii}^{\mathbf{g}\mathbf{0}\mathbf{0}\mathbf{0}}$ decays fast enough to prevail over a possibly substantial product of Green's functions. Specifically, for $R = 0.75 \text{ \AA}$, the largest product of the Green's functions corresponds to $|G_{ii}^{0\mathbf{0}}(\tau) G_{ii}^{0\mathbf{0}}(\tau) G_{ii}^{0\mathbf{0}}(-\tau)|$ at $\tau = 0$, and the integral $v_{1111}^{0\mathbf{g}\mathbf{g}\mathbf{g}\mathbf{g}}$ drops to the order of 10^{-6} for \mathbf{g} being 5 cells apart from the reference one.

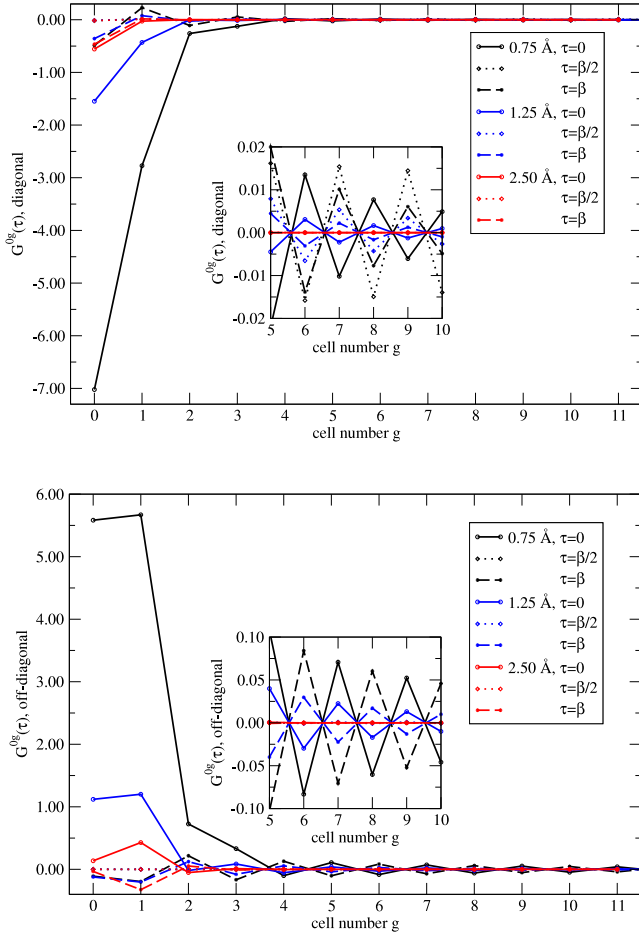


FIG. 1. The behavior of the Green's function matrix elements $G_{ij}^{0g}(\tau)$ (diagonal for $i = j = 1$ and off-diagonal for $i = 1, j = 2$) for the 1D hydrogen lattices with respect to the number of cells g separating the centers of basis functions i and j . Plotted for $R = 0.75, 1.25,$ and 2.50 \AA and $\tau = 0, \beta/2,$ and β for $\beta = 100$. Mini-Huzinaga basis⁶⁰ is used.

We observe that even the largest possible elements of $G(\tau)$ or v are contracted to facilitate a rapid decay of $\Sigma(\tau)$ with respect to the distance between cells $\mathbf{0}$ and \mathbf{g} . This local behavior of the self-energy is illustrated in Fig. 2. In agreement with the preceding discussion, at a moderate separation between the basis functions, the self-energy elements reduce to 10^{-5} – 10^{-4} in absolute value which supports the idea of the self-energy being relatively local, contrary, for example, to the density or Fock matrices. Consequently, despite significant $O(N^5)$ scaling of the self-energy evaluation, the blocks of $\Sigma_{ij}^{0g}(\tau)$ can be calculated for a rather moderate span of \mathbf{g} . The resulting $\Sigma_{ij}^{0g}(\tau)$ can be finally Fourier transformed to $\Sigma_{ij}^{0g}(\omega)$ to solve the Dyson equation. The transformations of the Green's function and self-energy between the imaginary-frequency and imaginary-time domains are made computationally efficient via a representation of the imaginary-time quantities in the basis of orthogonal (Legendre) polynomials.^{61,62}

IV. SELF-CONSISTENT EVALUATION OF THE 2ND-ORDER GREEN'S FUNCTION

At this point, we can assemble the components described in Sec. III in a self-consistent procedure for the crystalline

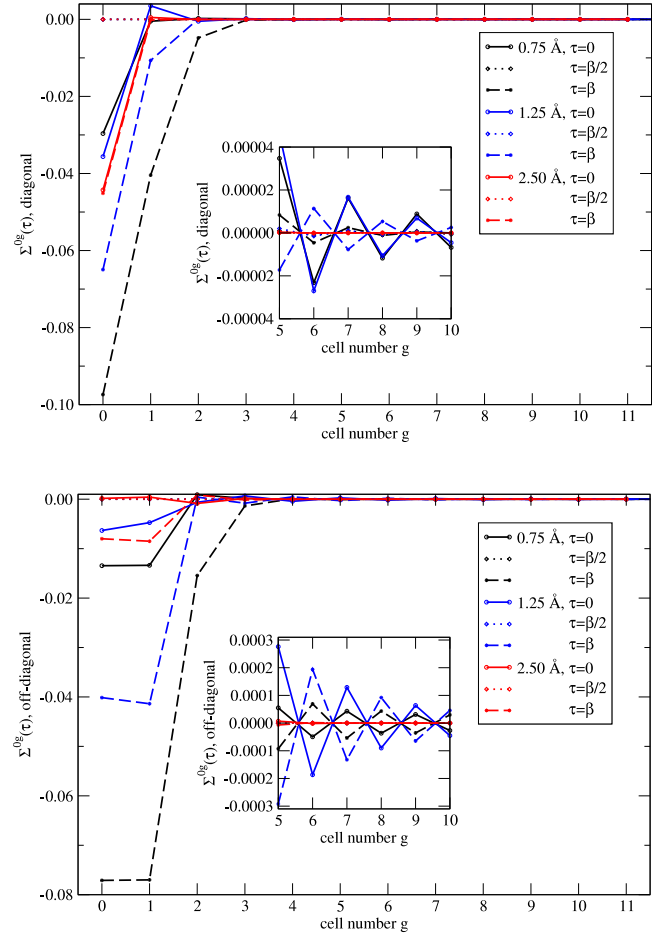


FIG. 2. The behavior of the self-energy matrix elements $\Sigma_{ij}^{0g}(\tau)$ (diagonal for $i = j = 1$ and off-diagonal for $i = 1, j = 2$) for the 1D hydrogen lattices with respect to the number of cells g separating the centers of basis functions i and j . Plotted for $R = 0.75, 1.25,$ and 2.50 \AA and $\tau = 0, \beta/2,$ and β ; same β and basis as in Fig. 1. In comparison to the Green's function, the corresponding elements of the self-energy approach zero much faster with the increase of g .

Green's function with the 2nd-order self-energy. The self-consistent GF2 procedure within the PBC framework, regardless of a particular implementation, has the same basic ingredients as the molecular one⁵⁵ and can be abstractly viewed as a double loop (Fig. 3).

- In the greater loop, the frequency-dependent 2nd-order self-energy is evaluated from a given Green's function and two-electron integrals, according to Eq. (8). The procedure is initiated with $\mathbf{G}_0^{\mathbf{k}}(\omega) = [(\omega + \mu)\mathbf{S}^{\mathbf{k}} - \mathbf{F}^{\mathbf{k}}]^{-1}$ where $\Sigma(\omega) = 0$ and $\mathbf{F}^{\mathbf{k}}$ typically comes from a converged HF or DFT calculation. Since the Green's function entering the loop is in the \mathbf{k} -space, an extra step $\mathbf{G}^{\mathbf{k}}(\omega)$ (from Eq. (5)) $\xrightarrow{\text{Eq. (6)}} \mathbf{G}^{0g}$ is necessary, compared to the molecular case. Most generally, once $\Sigma^{0g}(\tau)$ is evaluated and Fourier transformed to $\Sigma^{0g}(\omega)$, it should undergo the transformation $\Sigma^{0g}(\omega) \xrightarrow{\text{Eq. (4)}} \Sigma^{\mathbf{k}}(\omega)$ to enter a new Green's function evaluation at the common point of the two loops.
- In the lesser loop, the frequency-independent term Σ_{∞} is evaluated.

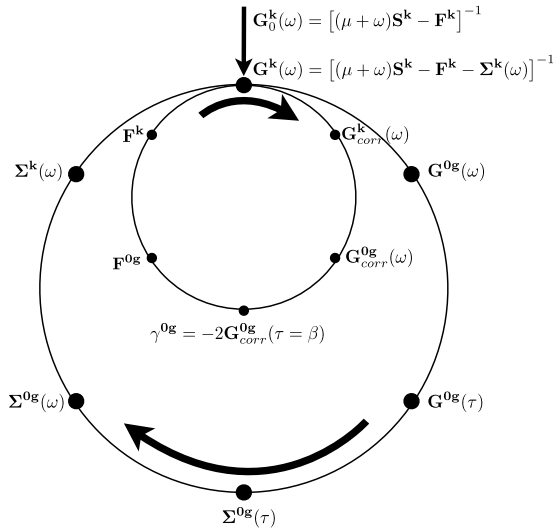


FIG. 3. GF2 self-consistency cycle.

The evaluation of Σ_∞ can be cast as constructing a Fock-type matrix $\mathbf{F} = \mathbf{h} + \mathbf{J}(\gamma) - \frac{1}{2}\mathbf{K}(\gamma) = \mathbf{h} + \Sigma_\infty$ from a correlated density matrix $\gamma = \gamma_{corr}$, where \mathbf{h} is the core Hamiltonian, $\mathbf{J}(\gamma)$ and $\mathbf{K}(\gamma)$ are the Coulomb and exchange operators, respectively. In practical realization of the method, any quantum chemistry software capable of treating crystalline problems can be used to take the correlated density as input and perform a single iteration (i.e., without reaching the self-consistency) of the regular HF procedure to build \mathbf{F} from a given γ . The correlated density matrix is evaluated as $\gamma_{corr} = -2\mathbf{G}_{corr}(\tau = \beta)$. To this end, the correlated Green's function $\mathbf{G}_{corr}^k(\omega)$ in the \mathbf{k} -space from Eq. (5) is transformed to the real space $\mathbf{G}_{corr}^{0g}(\omega)$ via Eq. (6) and then to the imaginary-time domain at $\tau = \beta$.

It is essential to assure that the resulting correlated density yields a correct number of electrons per unit cell. This is enforced by the chemical potential μ search: the value of μ is adjusted in such way that the correlated density matrix $\gamma_{corr}^{0g} = -2\mathbf{G}_{corr}^{0g}(\tau = \beta)$ contracted with the overlap matrix preserves the correct number of electrons in the unit cell $N_e = \sum_{\mathbf{g},i,j} \gamma_{ij}^{0g} \cdot \mathbf{S}_{ij}^{0g}$.

Now, $\mathbf{F} = \mathbf{h} + \Sigma_\infty$ can be evaluated as a single iteration of the HF procedure where a correlated density matrix is supplied to be contracted with the two-electron integrals. To close the lesser loop, one transforms \mathbf{F}^{0g} to the \mathbf{k} -space. At this point, all components are ready to continue to the greater loop until convergence is reached.

The change in the electronic energy per unit cell during iterations is used as a convergence criterion. This energy is a sum of two terms resulting from the frequency-independent and frequency-dependent parts of the self-energy. The frequency-independent part yields the ‘‘one-body’’ energy evaluated as

$$E_{1b} = \frac{1}{2} \sum_{\mathbf{g},i,j} \gamma_{ij}^{0g} (2\mathbf{h}_{ij}^{0g} + [\Sigma_\infty]_{ij}^{0g}) \quad (13)$$

using the *correlated* density matrix. The frequency-dependent part results in the ‘‘two-body’’ contribution

$$E_{2b} = \frac{2}{\beta} \sum_{\mathbf{g},i,j} \text{Re} \left[\sum_{\omega} \mathbf{G}_{ij}^{0g}(\omega) \Sigma_{ij}^{0g}(\omega) \right], \quad (14)$$

where $\mathbf{G}_{ij}^{0g}(\omega)$ is the *correlated* Green's function.

Let us mention that using the machinery we have just presented, one can also readily access temperature-dependent MP2 energy for a periodic system. According to Ref. 63, the MP2 correlation energy per unit cell is

$$E_{2b,MP2} = \frac{1}{\beta} \sum_{\mathbf{g},i,j} \text{Re} \left[\sum_{\omega} [\mathbf{G}_0(\omega)]_{ij}^{0g} \Sigma_{ij}^{0g}(\omega) \right], \quad (15)$$

where $\mathbf{G}_0(\omega)$ results from the HF Green's function $\mathbf{G}_0^k(\omega) = [(\omega + \mu)\mathbf{S}^k - \mathbf{F}^k]^{-1}$, and the self-energy is evaluated in a single run of the GF2 cycle using this starting HF Green's function $\mathbf{G}_0(\omega)$. The evaluation of the MP2 energy is not a self-consistent procedure, and the total energy comes out as $E_{HF} + E_{2b,MP2}$.

V. EVALUATION OF SPECTRAL FUNCTIONS

Matsubara Green's functions, in contrast to the real-axis ones, are smooth and convenient to use in iterative schemes, but they do not provide a direct access to the spectral properties of a system. Analytic properties of the Matsubara Green's function on the imaginary axis assure the existence and uniqueness of its analytical continuation to the real frequency axis. Numerical analytical continuation of the Matsubara data is an ill-posed problem,⁶⁴ but it is required for obtaining spectral functions from imaginary-time algorithms.^{65,66}

In this work, we use the maximum entropy analytical continuation method as implemented in Ref. 67 to obtain the spectral function. The spectral function is defined as

$$A(\mathbf{k}, \omega) = -\frac{1}{\pi} \text{Tr} [\text{Im} \mathbf{G}^k(\omega) \mathbf{S}^k], \quad (16)$$

where $\mathbf{G}^k(\omega)$ is the real-frequency Green's function, and \mathbf{S}^k is the overlap matrix introduced to generalize the expression to a non-orthogonal basis. The spectral function $A(\mathbf{k}, \omega)$ is a momentum-dependent density of states and gives the number of states available to the electrons with energy ω and momentum \mathbf{k} .

The success of constructing the spectral function can be verified via the ‘‘back continuation’’ to the imaginary axis by calculating the following integral:

$$G^k(\omega_n) = \int_{-\infty}^{+\infty} d\omega \frac{A(\mathbf{k}, \omega)}{\omega_n - \omega}, \quad (17)$$

where ω_n stands for the n th Matsubara frequency.

VI. ELECTRON CORRELATIONS IN ONE-DIMENSIONAL PERIODIC HYDROGEN

1D periodic hydrogen is the simplest crystalline system described by a realistic Hamiltonian which makes it suitable for testing the periodic GF2 method. Despite its simplicity, depending on the interatomic distances (corresponding to different pressures) such system displays three phases: a metal, a band insulator, and a Mott insulator.

In this study, we employ the mini-Huzinaga basis set⁶⁰ that has one s -function per hydrogen atom. To remain within the spin-restricted formulation of GF2, we choose two hydrogen atoms per unit cell and assume a closed-shell one-electron reference throughout this discussion. In all Green’s function evaluations, we use 5000 Matsubara frequencies and the inverse temperature $\beta = 100$ a.u.⁻¹ (corresponding to 3157.8 K). The update of the frequency-independent part of the self-energy present in the GF2 method is executed via running a single iteration of HF using the GAUSSIAN⁶⁸ program that takes the correlated density matrix as input.

A. Convergence of the electronic energy

The GF2 energy is calculated as $E_{1b} + E_{2b}$, where E_{1b} and E_{2b} are defined in Eqs. (13) and (14). The one-body energy (Eq. (13)) is evaluated after the Fock matrix is created using the correlated density matrix; the cell index \mathbf{g} runs over the entire crystal. Similarly, the two-body energy, E_{2b} (Eq. (14)), formally requires a summation over the infinite number of cell indices \mathbf{g} . In Sec. III B, we have illustrated that the frequency-dependent self-energy is quite local. Therefore, it is natural to expect that a relatively moderate span of cell indices \mathbf{g} is required to converge the two-body energy sum in Eq. (14). Since the evaluation of $\Sigma(\tau)$ is the computational bottleneck of the GF2 calculation, a small range of the cell indices \mathbf{g} is crucial for making our calculations computationally affordable.

To assess the effect of truncating the summation, we evaluate the energy from Eq. (14) as a function of the number of cells. In Table I, we illustrate the unit cell E_{2b} behavior for three spacings of the lattice from tight (0.75 Å) to relatively wide (2.50 Å). The presented values correspond to a converged GF2 iterative procedure with the GF2 energy convergence criterion 10^{-6} a.u.

Even for the most tight spacing $R = 0.75$ Å, E_{2b} reaches the 10^{-3} a.u. plateau for 3 cells and 10^{-5} a.u. for 11 cells, while the same convergence criteria for the E_{1b} term are only met for at least twice as many cells. Thus, while with the increase of the intercell distance the number of cells required to converge E_{2b} rapidly drops, the number of cells necessary to converge E_{1b} decreases quite slowly: for $R = 0.75, 1.25,$ and 2.50 Å converging E_{1b} to 10^{-5} a.u. requires, respectively,

TABLE I. The convergence of E_{2b} , a.u., per unit cell with respect to $\#N$ — the number of unit cells used in Eq. (14).

#N	R, Å		
	0.75	1.25	2.50
1	-0.205 953	-0.104 473	-0.127 235
2	-0.120 203	-0.078 355	-0.128 165
3	-0.034 454	-0.052 237	-0.129 095
4	-0.035 047	-0.052 109	-0.129 024
5	-0.035 640	-0.051 980	-0.128 953
6	-0.035 553	-0.051 918	-0.128 949
7	-0.035 466	-0.051 855	-0.128 944
8	-0.035 478	-0.051 830	-0.128 944
9	-0.035 490	-0.051 805	-0.128 943
10	-0.035 495	-0.051 793	-0.128 943
11	-0.035 500	-0.051 781	-0.128 943

25, 21, and 17 unit cells. Although no universal recipe can be derived from this simple case, the convergence rate of the E_{2b} part of the GF2 energy with respect to the number of cells explicitly summed over in Eq. (8) is fast due to the local nature of the 2nd-order self-energy and can lead to computational time savings.

B. 1D equidistant hydrogen energy curve

We examine an equidistant stretching of the 1D periodic hydrogen lattice. HF and MP2 energy curves in Fig. 4 display features that are qualitatively similar to those previously observed in Refs. 55 and 63. Specifically, closed-shell HF significantly overestimates the unit cell energy at dissociation, while MP2 diverges for interatomic distances exceeding 2 Å.

In an equidistant stretch of the 1D periodic hydrogen lattice, multiple phases with different Helmholtz free energies can be present. A GF2 calculation, where the Green’s function is evaluated on the Matsubara axis, yields the electronic part of the internal energy. However, since GF2 is a nonlinear iterative procedure, for a given crystalline geometry it can converge to different energy solutions depending on the starting point. The existence of multiple solutions in nonlinear iterative methods is not uncommon and is known to transpire in the HF⁶⁹ and GW^{70,71} methods. Thus, to determine the most stable phase based on Helmholtz free energy, the electronic entropy contribution needs to be evaluated. We will refrain from evaluating the entropic contribution in this publication and address it separately. Here, we will focus on analyzing possible phases and phase transitions based solely on the electronic internal energy.

Although multiple solutions are possible, in this particular calculation, we have observed convergence to two solutions. The green curve in Fig. 4 (“solution 1”) corresponds to the solution obtained using the HF initial guess. Superficially, this curve resembles the behavior of the finite species in Ref. 55. The blue curve in Fig. 4 (“solution 2”) is obtained from a starting point that assumes no interactions between hydrogen atoms (“atomic” initial \mathbf{F}^k matrices). In the region close to equilibrium, both GF2 solutions are practically

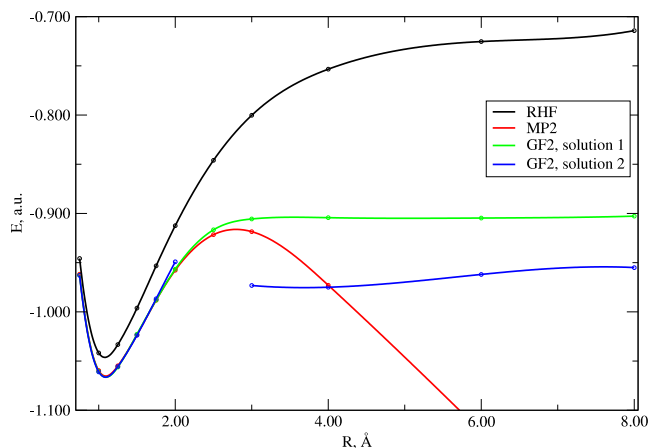


FIG. 4. Equidistant 1D hydrogen stretching curves: electronic energies per unit cell for restricted closed-shell Hartree–Fock (RHF), 2nd-order Møller–Plesset perturbation theory (MP2), and two GF2 solutions.

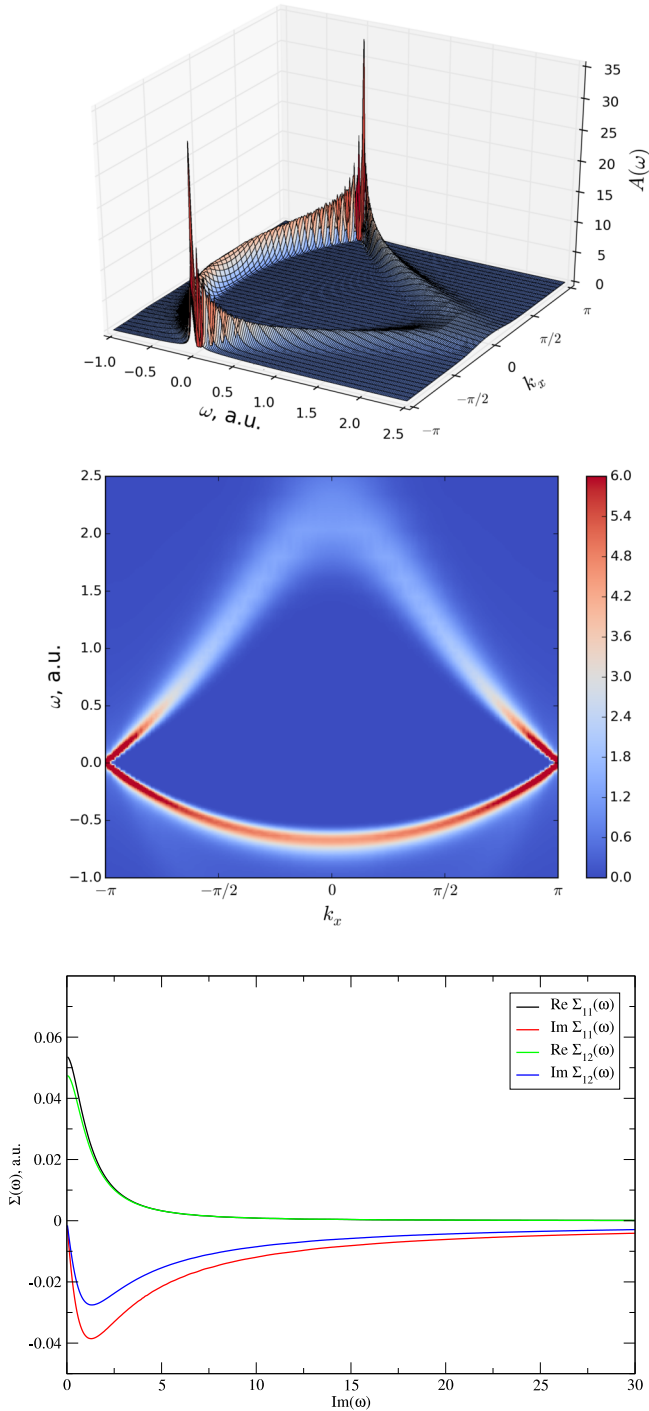


FIG. 5. Real-frequency spectral function and correlated bands, Matsubara self-energy for “solution 1” at $R = 0.75 \text{ \AA}$.

indiscernible from MP2. Once MP2 breaks down around past 2 \AA , the blue curve (“solution 2”) separates from the green one (“solution 1”) and stays consistently lower in energy until complete atomization. We observe that around 2 \AA , the blue “solution 2” undergoes a phase transition, and we experience difficulties converging GF2 near the geometry where it happens (this explains a blank region on the blue curve around 2.5 \AA).

In Sec. VI D, based on occupation numbers we will show that both GF2 solutions undergo phase transitions during the equidistant stretch. Since the metallic, band insulating, and

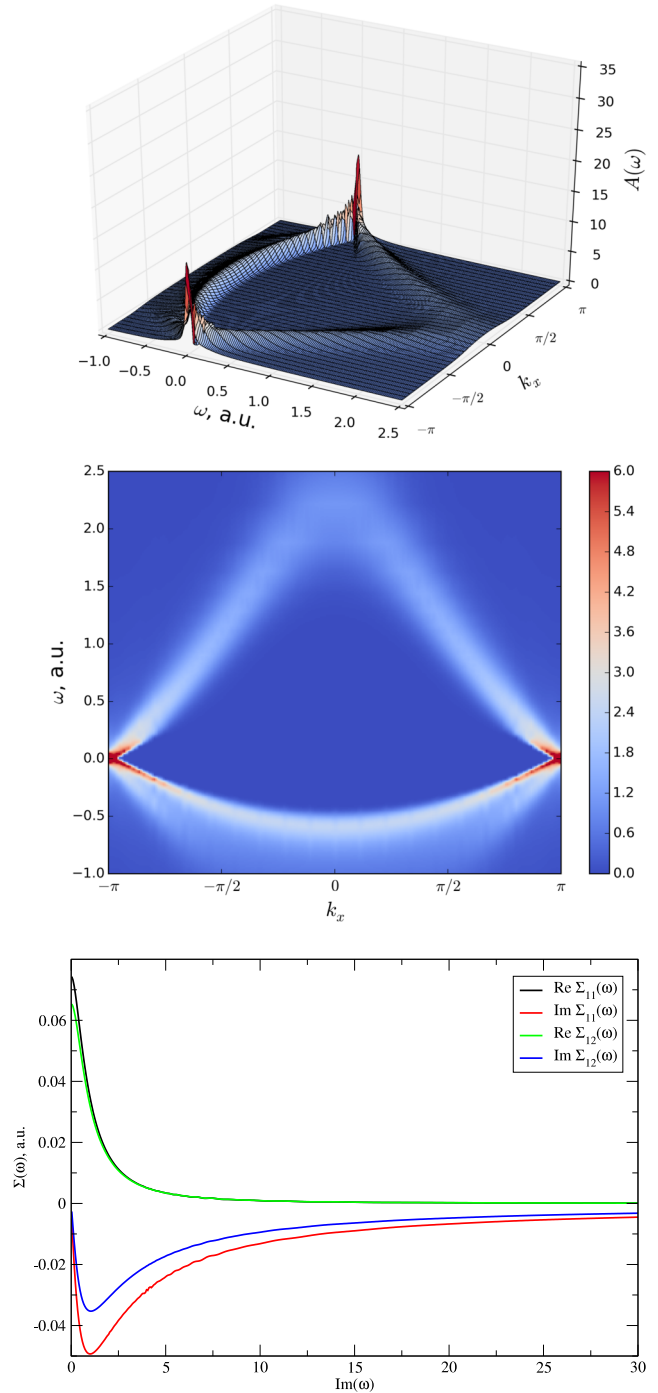


FIG. 6. Real-frequency spectral function and correlated bands, Matsubara self-energy for “solution 2” at $R = 0.75 \text{ \AA}$.

Mott insulating character of a solution cannot be concluded from the energy profile alone, we will turn to analyzing spectral functions, self-energies, and occupation numbers for “solution 1” and “solution 2.”

C. Self-energies and spectral functions of the 1D hydrogen lattice

We present the self-energies and spectral functions since these behave differently for different phases and provide a unique characterization of a solution.

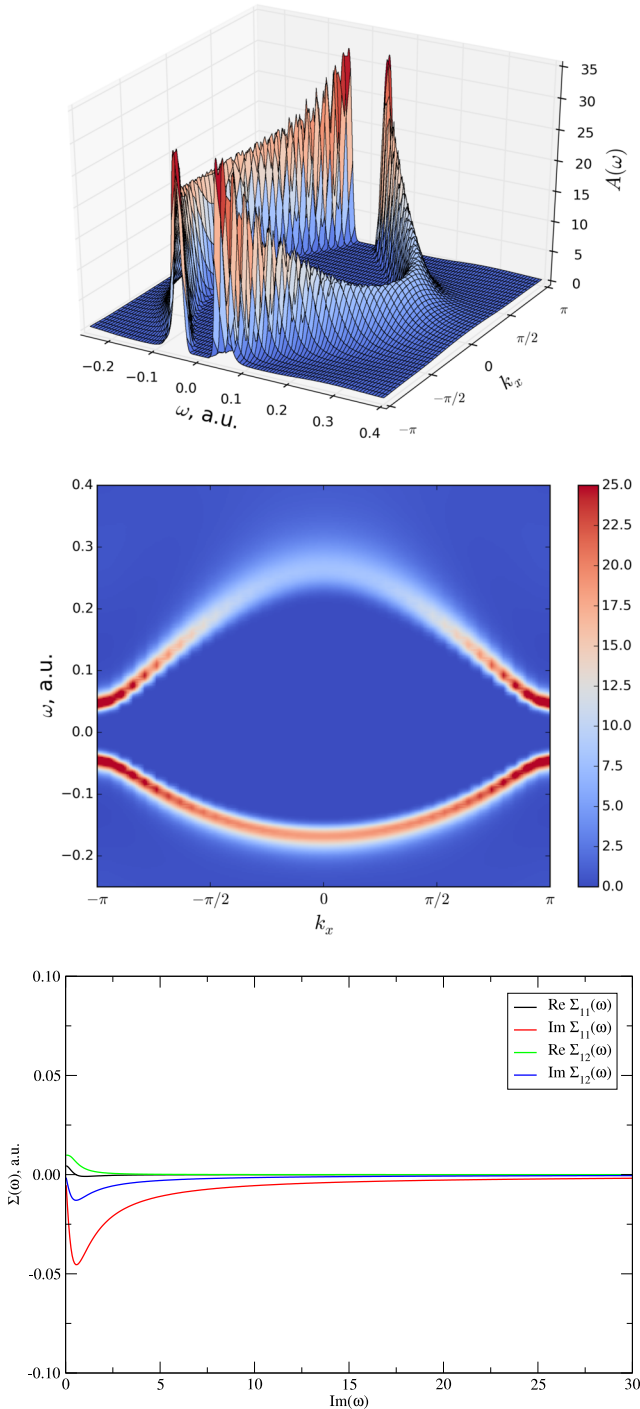


FIG. 7. Real-frequency spectral function and correlated bands, Matsubara self-energy for “solution 1” at $R = 1.75 \text{ \AA}$.

We plot the angle-resolved spectral function $A(\mathbf{k}, \omega)$ both as a 3D plot and a 2D projection on the k - ω plane using matplotlib.⁷² The color of the projection indicates the “height” of the original spectral function. These projections can be interpreted as remnants of the bands similar to the ones resulting from one-body methods such as HF or DFT. However, it is important to note that these correlated bands do not have the same meaning as bands in the one-electron picture since they emerge from a many-body calculation.

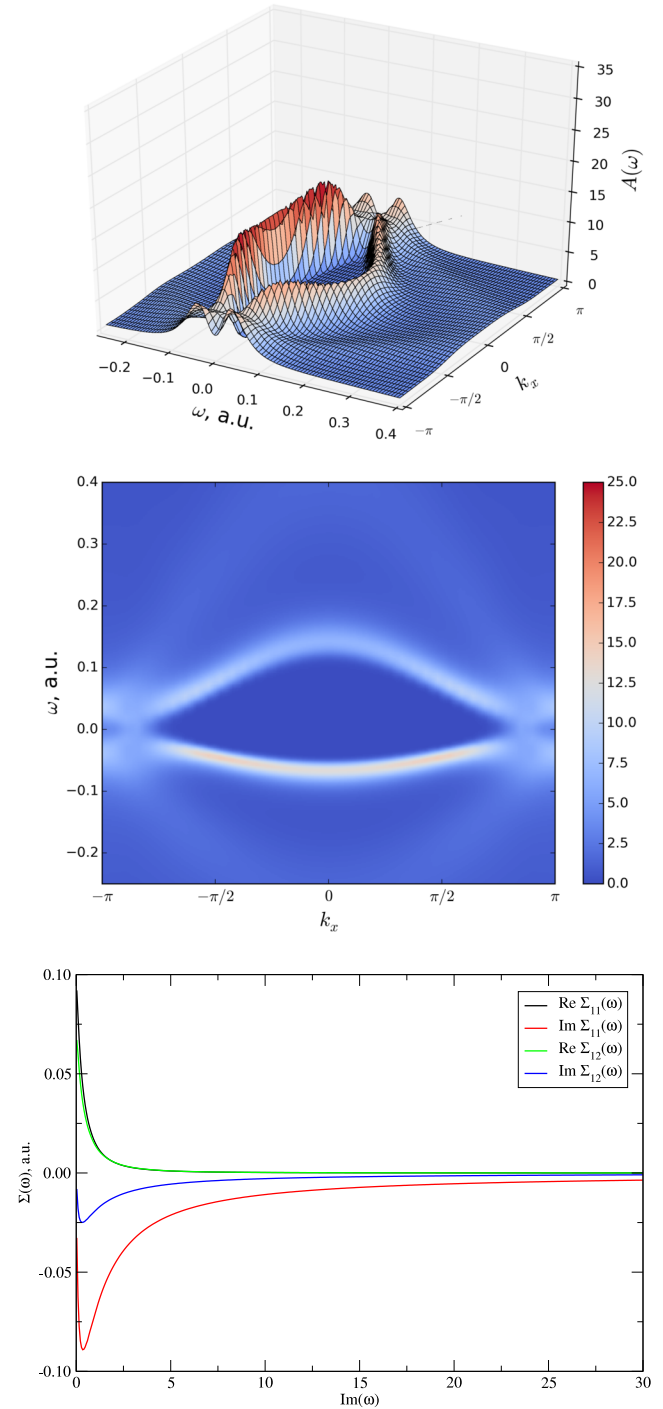


FIG. 8. Real-frequency spectral function and correlated bands, Matsubara self-energy for “solution 2” at $R = 1.75 \text{ \AA}$.

1. Short bond length/high pressure

In the condensed regime, $R = 0.75 \text{ \AA}$, the green curve (“solution 1”) displays metallic behavior with two peaks of the spectral function $A(\mathbf{k}, \omega)$ at $k = \pm\pi$ for $\omega = 0$, as can be seen from Fig. 5. The resulting spectral function projection shows that the band gap closes at $k = \pm\pi$ as expected for a metallic solution. The Matsubara self-energies (the third panel in Fig. 5 contains the diagonal $\Sigma_{11}(\omega)$ and off-diagonal $\Sigma_{12}(\omega)$ elements, where ω 's are

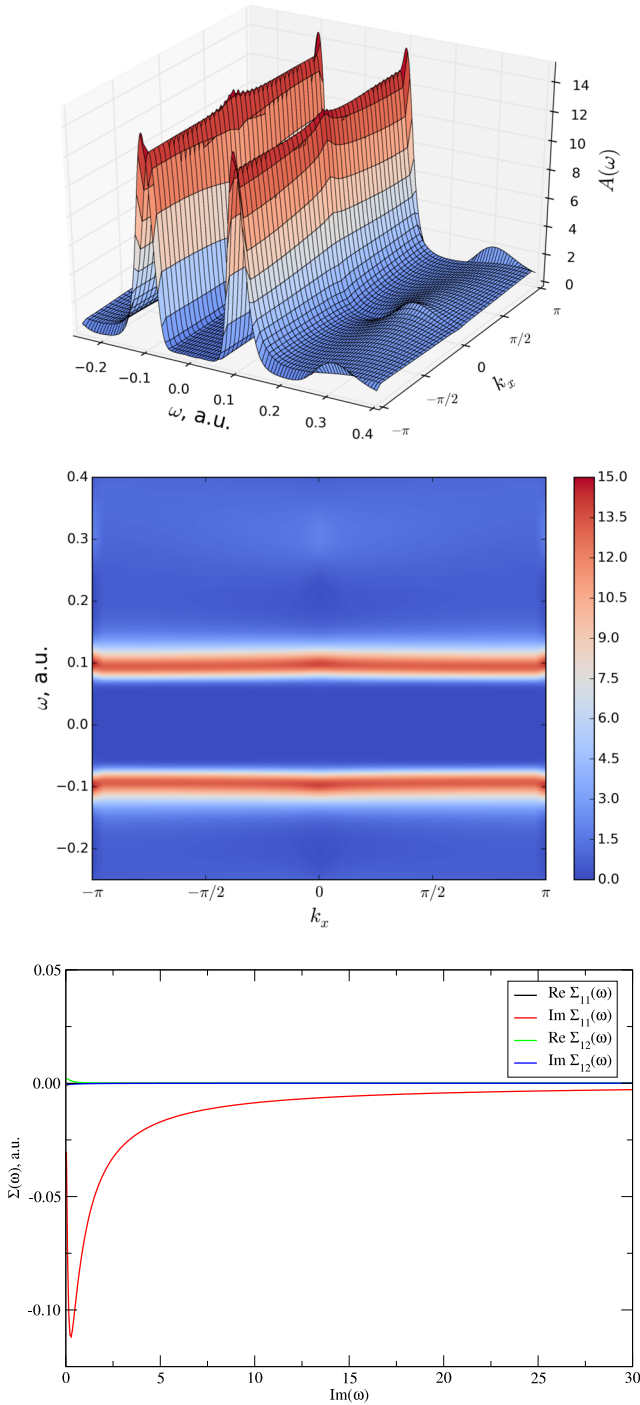


FIG. 9. Real-frequency spectral function and correlated bands, Matsubara self-energy for “solution 1” at $R = 4.00 \text{ \AA}$.

imaginary frequencies $\omega_n = i(2n + 1)\pi/\beta$) for “solution 1” display a typical Fermi liquid behavior that is characteristic for both metals and band insulators. The metallic behavior of “solution 1” is confirmed by the spectral function showing a non-zero density of states at the Fermi energy.

The other GF2 solution, “solution 2,” for $R = 0.75 \text{ \AA}$ illustrated in Fig. 6, also displays a metallic character and shows no gap for $k = \pm\pi$ at $\omega = 0$. However, this spectral function looks different than the one for “solution 1.” The self-energy for “solution 2” also has a Fermi liquid character with the imaginary part very similar to “solution 1,” while

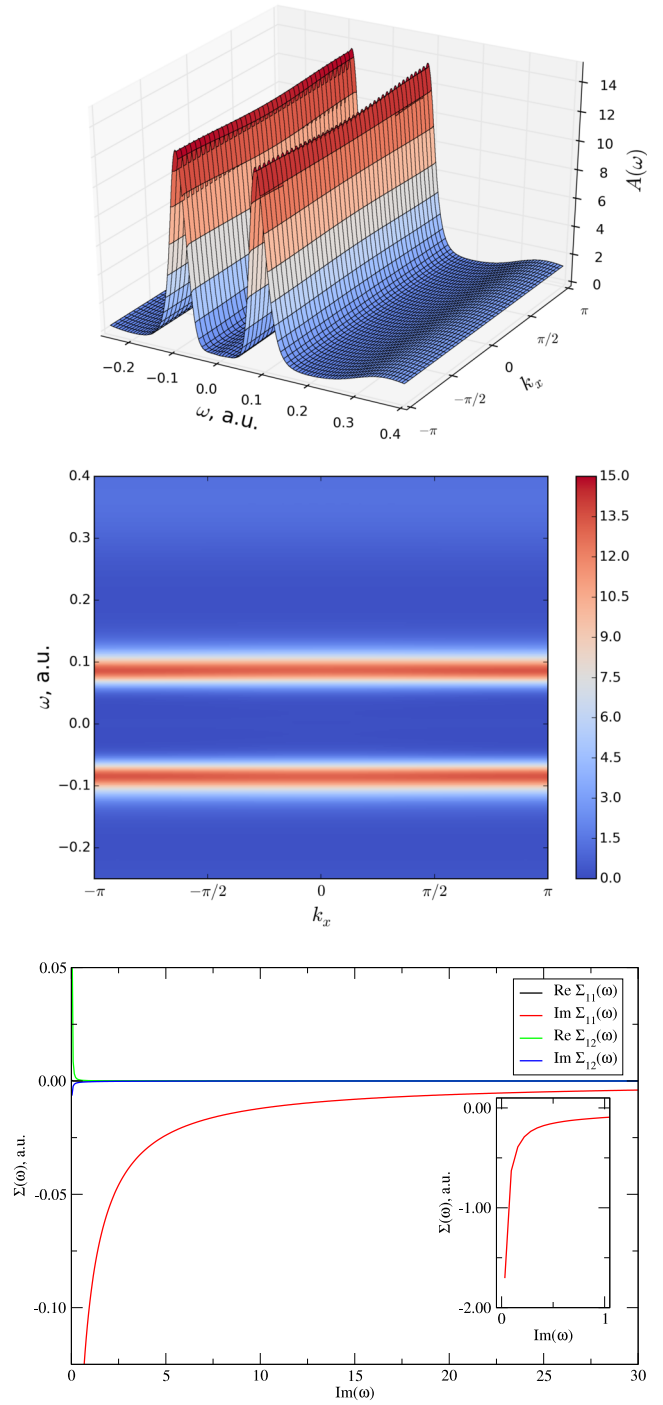


FIG. 10. Real-frequency spectral function and correlated bands, Matsubara self-energy for “solution 2” at $R = 4.00 \text{ \AA}$.

small differences between the two solutions are visible in the real part. As we have mentioned previously, an entropy contribution is necessary to determine which of the two solutions gives rise to the most stable metallic phase at a given temperature.

2. Intermediate bond length/intermediate pressure

The intermediate bond length regime, $R = 1.75\text{--}2.00 \text{ \AA}$, is very interesting since both solutions display qualitatively different behaviors. As illustrated in Fig. 7, the spectral

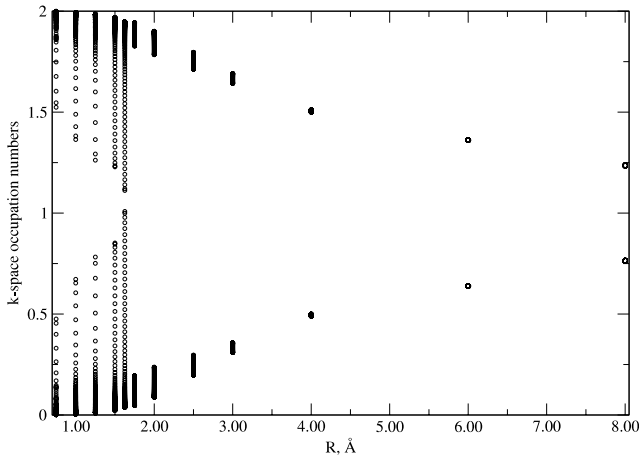


FIG. 11. Equidistant 1D hydrogen: \mathbf{k} -space natural occupations for “solution 1.”

function of “solution 1” clearly shows an insulating behavior with two bands separated by a band gap. The self-energy of “solution 1” is not divergent near zero frequency; consequently, the phase that emerges is a band insulator and not a Mott insulator.

The spectral function of “solution 2” in Fig. 8 is gapless for $\omega = 0$ indicating a metallic phase. The self-energy of “solution 2” has a Fermi liquid profile and confirms the metallic character of this solution.

3. Long bond length/low pressure

In the stretched regime, $R = 4.00 \text{ \AA}$, both solutions correspond to insulators. The spectral function of “solution 1,” illustrated in Fig. 9, displays an insulating behavior and its self-energy confirms its band insulator character.

The spectral function of “solution 2” in Fig. 10 has two sharp peaks near $\omega = -0.1$ and 0.1 a.u. The self-energy of “solution 2” is large for the frequencies near $\omega = 0$ confirming a Mott insulator behavior. A Mott insulator phase for 1D periodic hydrogen lattice can be described by an open-shell singlet wavefunction encompassing all hydrogen atoms in the crystal.

Let us stress here that recovering the Mott behavior requires accounting for the multi-reference character of the electronic state. Consequently, a zero-temperature perturbation theory such as MP2 would experience divergencies and would not be able to illustrate the Mott phase. The GF2 success in qualitatively capturing this phase stems from its iterative character and inclusion of higher-order diagrams than the ones originally included in the 2nd-order self-energy expansion. GF2 recovers the Mott solution only qualitatively and can be used as a first step in embedding schemes such as self-energy embedding theory^{73,74} if quantitative results are desired. Other theories such as density matrix renormalization group (DMRG)⁷⁵ and quantum Monte-Carlo (QMC)⁷⁶ that are more expensive than GF2 can recover the Mott regime quantitatively.

D. Occupation numbers in the \mathbf{k} -space

Natural occupation numbers, i.e., the eigenvalues of the density matrix, provide an additional insight into electronic

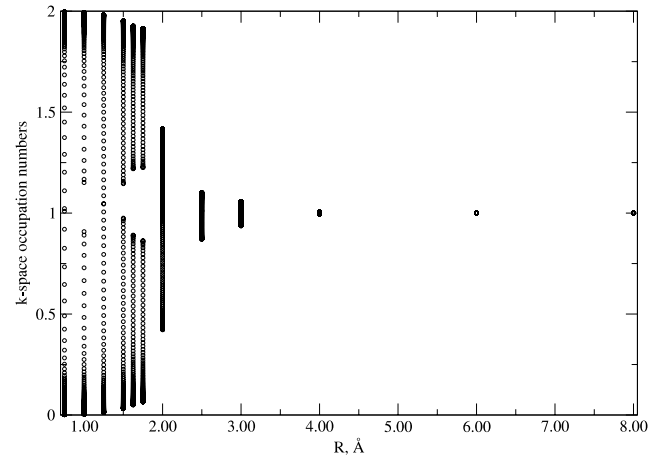


FIG. 12. Equidistant 1D hydrogen: \mathbf{k} -space natural occupations for “solution 2.”

correlations and phase transitions present in the system. Since we work in the non-orthogonal AO basis, we calculate the density matrix in the Löwdin orthogonalized basis at every \mathbf{k} ,

$$\gamma_{orth}^{\mathbf{k}} = (\mathbf{S}^{\mathbf{k}})^{-1/2} \gamma^{\mathbf{k}} (\mathbf{S}^{\mathbf{k}\dagger})^{-1/2}, \quad (18)$$

and diagonalize it to obtain natural \mathbf{k} -space occupation numbers. Generally, different numbers of \mathbf{k} -points are required to achieve convergence at different geometries, and the number of the density matrix eigenvalues changes with the interatomic distance.

The eigenvalues of $\gamma_{orth}^{\mathbf{k}}$ along the “solution 1” curve are plotted in Fig. 11 for all momentum vectors \mathbf{k} used to sample the first Brillouin zone. The behavior of these natural occupations starkly contrasts the finite system behavior analyzed in Ref. 55. In the short bond regime ($0.75\text{--}1.75 \text{ \AA}$), the infinite 1D hydrogen lattice features a much wider spectrum of fractional occupations. Thus, even around equilibrium, the ground state of the 1D infinite hydrogen lattice acquires some multi-reference character. At $R = 1.75 \text{ \AA}$, the natural occupations discontinuously change from a wide spectrum to two relatively narrow clusters in the vicinity of 0 and 2 indicating a possible phase transition. Upon further stretching, these clusters shrink and depart from 0 and 2 occupations, but never collapse to 1 that would be indicative of a Mott regime.

The occupation numbers in Fig. 12 corresponding to “solution 2” (blue curve in Fig. 4) also display a multi-reference nature of the electronic state around equilibrium. In contrast to “solution 1,” however, for interatomic distances beyond 2 \AA , the occupation spectrum rapidly collapses to 1 indicating a clear phase transition to the strongly correlated Mott phase.

E. Peierls distortion of the 1D infinite hydrogen lattice

Equidistant 1D hydrogen lattices are known to undergo the Peierls distortion,⁷⁷ i.e., they energetically favor an alternant pattern of bond lengths to an equidistant one. In Fig. 13, we plot the energies and natural occupation numbers for the 1.50 \AA equidistant 1D hydrogen lattice undergoing a slight distortion resulting in an interatomic distance pattern $R + \delta R - R - \delta R$. A small alteration of the interatomic separation pattern by $\delta R = 0.05 \text{ \AA}$ results in a discontinuous

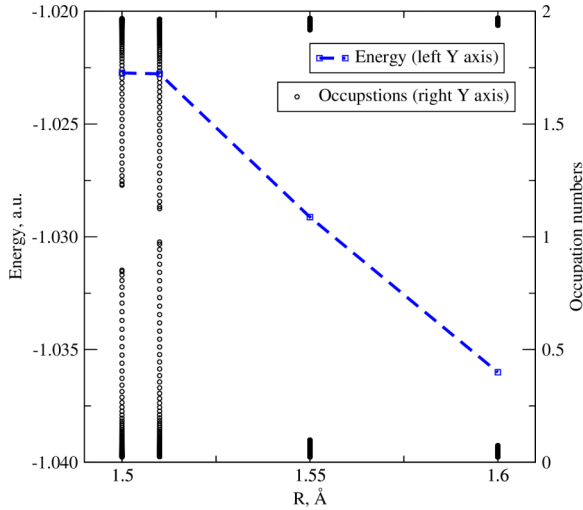


FIG. 13. Energies and \mathbf{k} -space natural occupations under the Peierls distortion of the 1.50 Å equidistant 1D hydrogen lattice.

change in the natural \mathbf{k} -occupations, indicative of a phase transition, accompanying the energy decrease.

Indeed, while the occupation numbers of equidistant 1D hydrogen lattices in Figs. 11 and 12 display a noticeable multi-reference character for distances below 2 Å, the $\delta R = 0.05$ Å distorted lattice is quite weakly correlated and has occupations 0 and 2 for distances greater than 1.52 Å. Moreover, the appearance of the spectral functions (Fig. 14) at $k = -\pi$ for several δR clearly indicates a metal to band insulator transition under such distortion.

F. The importance of GF2 iterations

In Table II, we list the unit cell energy and density matrix elements for the $R = 0.75$ Å equidistant lattice in the course of GF2 iterations. The essential observation is that the convergence is not reached in a single iteration, but it takes several GF2 cycles before the unit cell energy and density matrix elements stabilize.

To further highlight the importance of executing the GF2 cycle in a self-consistent fashion, updating both the frequency-

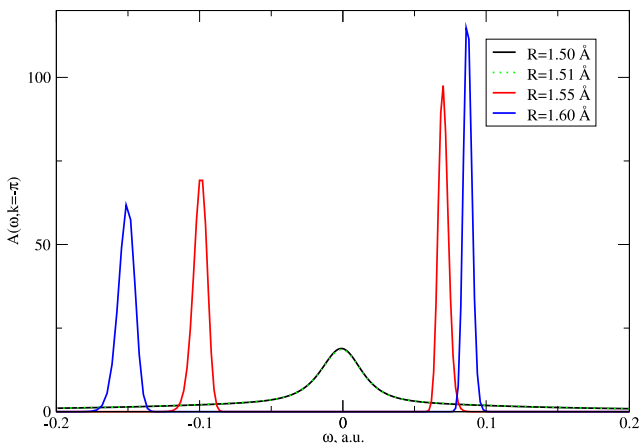


FIG. 14. Gap opening in 1D hydrogen at $R = 1.50 + \delta R$ Å.

TABLE II. Equidistant 1D hydrogen lattice, $R = 0.75$ Å, 2 atoms per unit cell: reference cell energy and density matrix elements in the course of GF2 iterations.

Iteration	E, a.u.	γ_{11}^{00}	γ_{12}^{00}
Hartree-Fock	-0.944 77	0.905	0.358
First	-0.962 02	0.954	0.520
Last (converged)	-0.962 72	0.954	0.251

dependent and frequency-independent parts of the self-energy, we consider the behavior of the spectral function of the same lattice in the course of the iterations. Such lattice is metallic at the convergence of iterations for both GF2 solutions. In Fig. 15, we show the “slices” of the spectral functions at $k = -\pi$ corresponding to the HF reference, single iteration of GF2, correlated Green’s function at convergence containing the Σ_∞ term and neglecting the frequency-dependent term $\Sigma(\omega)$, and finally the fully converged GF2.

The metallic behavior transpires only at the level of the fully self-consistent GF2 solution with electron correlations incorporated in both the updated Σ_∞ term and the explicitly frequency-dependent $\Sigma(\omega)$ part of the self-energy. Whenever the full iterative cycle is not performed, qualitatively different gapped solutions appear.

G. Convergence of GF2 for metallic solutions

We briefly comment on the convergence of the GF2 method for metallic systems. Zero-temperature 2nd-order MP2 is known to diverge for metals with increasing number of \mathbf{k} -points as was theoretically demonstrated in Ref. 78, p. 32 (see also Refs. 79–81) and numerically observed by Hirata²⁸ and Kresse.³⁸ RPA does not encounter these divergencies^{81,82} due to the inclusion of an infinite series of different order “bubble” diagrams.

In contrast to MP2, in GF2 we do not observe any divergence for metallic solutions of the 1D hydrogen lattice. As shown in Table III for $R = 0.75$ Å, the total unit cell energy and the E_{2b} term (defined in Eq. (14)) are convergent with respect to the number of \mathbf{k} -points.

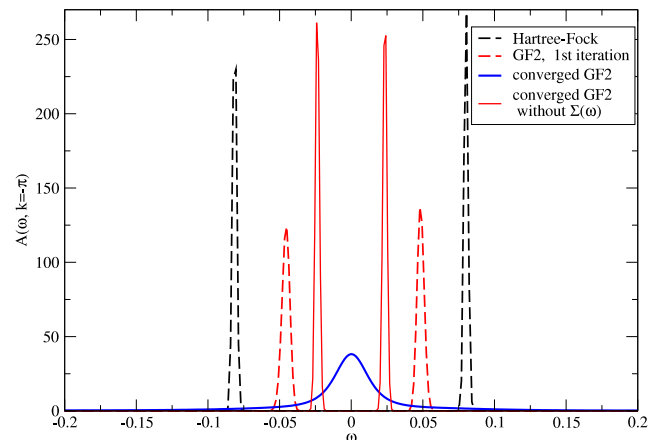


FIG. 15. $A(\omega, k = -\pi)$ at $R = 0.75$ Å with the inclusion of various elements of the GF2 procedure.

TABLE III. Unit cell total energies and 2-body electronic energies calculated using different numbers of \mathbf{k} -points for the metallic solution of the 1D equidistant periodic hydrogen, $R = 0.75 \text{ \AA}$. Given k 's correspond to the number of \mathbf{k} -points in the inversion-unique half of the first Brillouin zone; for each k there are $2k - 1$ points in the full first Brillouin zone. $k = 226$ is sufficient to attain (standard) GAUSSIAN convergence requirements.

Number of \mathbf{k} -points	Total energy	2-body energy
226	-0.962 719	-0.035 374
400	-0.962 685	-0.035 818
600	-0.962 715	-0.035 502
800	-0.962 715	-0.035 502

Several key differences to the MP2 case exist. The presence of only a single bubble diagram in traditional MP2 leads to divergencies in metals. While GF2 also contains a single bubble in the explicit self-energy diagrammatic expression, the GF2 iterations result in the inclusion of a series of increasingly higher-order diagrams. We surmise that in the self-consistent GF2, other bubble diagrams called clothed skeleton diagrams appear in the diagrammatic expansion. These diagrams, however, are different from an infinite sum of bubbles in RPA, as they appear due to renormalization of the Green's function lines. Consequently, we can refer to self-consistent GF2 as a renormalized perturbation theory. Its diagrammatic expansion can remain convergent and potentially cure the self-consistent GF2 of the MP2-type divergencies.

The latter statement is supported by earlier findings in the mathematical literature⁸³ and the work of Neumayr and Metzner⁸⁴ exploring Fermi systems within two-dimensional Hubbard model. In the latter publication, a renormalized second-order perturbation theory which resembles our GF2 scheme and contains a series of clothed diagrams is applied to different regimes. According to Ref. 84, "the basic idea is to choose an improved starting point for the perturbation expansion, by adding a suitable counterterm to the noninteracting part of the Hamiltonian, and subtracting it from the interaction part. The counterterm is quadratic in the Fermi operators and has to be determined from a self-consistency condition." This suggestion appears in the same vein as iterating the static and time-dependent terms of the self-energy until convergence in self-consistent GF2.

While the divergence of MP2 for metallic solutions at finite temperatures in 1D systems is established,²⁸ the convergent behavior of GF2 can be particular to 1D case. Therefore, further analysis of 2D and 3D systems is required.

VII. CONCLUSIONS

We have presented a prototype implementation of the finite-temperature iterative 2nd-order Green's function theory for periodic systems.

Our GF2 implementation is based on the evaluation of the self-energy in the real space using atomic orbitals and solving the Dyson equation in \mathbf{k} -space. Consequently, the resulting self-energy is local and decays fast with respect to the number of cells included. This fast decay of the AO self-energy can be used to increase computational robustness of our implementation. Solving the Dyson equation in the

reciprocal space via independent matrix inversions at every \mathbf{k} -point facilitates an efficient evaluation of the Green's function.

To learn about the performance of GF2 for solids, we have analyzed a 1D infinite hydrogen lattice. Our simulations at $T \approx 3160 \text{ K}$ ($\beta = 100 \text{ a.u.}^{-1}$) for each interatomic distance converge to two phases different in internal energy, spectral properties, or both.

We have confirmed that GF2 is capable of describing metallic phases of 1D hydrogen lattice, and it is not affected by the divergencies typical for the zero-temperature MP2.

GF2 is capable of qualitatively describing a metal-to-insulator transition. The resulting Mott phase that requires a multi-determinantal wavefunction can be qualitatively captured, and the 1D hydrogen lattice can be qualitatively correctly atomized. We attribute the emergence of the Mott phase in GF2 to the iterative nature of the method implicitly accounting for the multi-determinantal wavefunction via a self-consistent update of the Fock-type reference using the correlated density matrix. This appears sufficient to qualitatively cure the known deficiencies of single-reference methods, including MP2, in stretching and breaking bonds.

Our discernment of various phases of equidistant 1D hydrogen relies on a careful examination of the behavior of the natural occupations, self-energy, and spectral function. The \mathbf{k} -dependent spectral function obtained via the analytical continuation of the Matsubara Green's function to the real axis, coupled with the self-energy behavior analysis, allows us to recognize metallic, band insulator, and Mott phases.

To the best of our knowledge, GF2 is one of the very few available *ab initio* methods to account, at least qualitatively, for weak and strong electronic correlations simultaneously that can be applicable to metals, band, and Mott insulators. Therefore, a natural question arises if GF2 can be applied to large systems with multiple orbitals in the unit cells. The formal scaling of periodic GF2 is $O(N_{orb}^5 N_{cell}^4 n_\tau)$ where N_{orb} , N_{cell} , and n_τ are, respectively, the number of orbitals in the unit cell, the number of unit cells included in the evaluation of the self-energy, and the size of the imaginary-time grid. Though the calculation is easily parallelized over the imaginary-time points, it can be expected that GF2 is still more expensive than traditional MP2. For this reason, further development of GF2 applications to systems with large numbers of orbitals per unit cell can benefit from employing stochastic techniques to evaluate the 2nd-order self-energy.⁸⁵⁻⁹⁰ Efforts in this direction are underway.

ACKNOWLEDGMENTS

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